



Ferromagnetic ordering in ThSi₂ type CeAu_{0.28}Ge_{1.72}

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ABSTRACT

The compound CeAu_{0.28}Ge_{1.72} crystallizes in the ThSi₂ structure type in the tetragonal space group *I4₁/amd* with lattice parameters $a=b=4.2415(6)$ Å $c=14.640(3)$ Å. CeAu_{0.28}Ge_{1.72} is a polar intermetallic compound having a three-dimensional Ge/Au polyanion sub-network filled with Ce atoms. The magnetic susceptibility data show Curie–Weiss law behavior above 50 K. The compound orders ferromagnetically at ~8 K with estimated magnetic moment of 2.48 μ_B/Ce. The ferromagnetic ordering is confirmed by the heat capacity data which show a rise at ~8 K. The electronic specific heat coefficient (γ) value obtained from the paramagnetic temperature range 15–25 K is ~124(5) mJ/mol K². The entropy change due to the ferromagnetic transition is ~4.2 J/mol K which is appreciably reduced compared to the value of $R \ln(2)$ expected for a crystal-field-split doublet ground state and/or Kondo exchange interactions.

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1. Introduction

Intermetallic compounds of Ce exhibit a wide range of physical properties including magnetically ordered Kondo lattice behavior in CeRhSn₂ [1,2], heavy fermion behavior in CeCu₆ [3], superconductivity in CeCu₂Si₂ [4], non-fermi liquid behavior in CeNi₂Ge₂ [5], valence fluctuations in Ce₂Co₃Ge₅ [6], etc. These unusual properties are associated with the hybridization of the moment carried by the 4*f* electrons with the conduction electrons. Ce can exhibit two electronic configurations: the magnetic Ce³⁺ (4*f*¹) and non-magnetic Ce⁴⁺ (4*f*⁰). In most cases, the magnetic moments derive from Ce³⁺ and order antiferromagnetically at lower temperature. There are several examples, however, that exhibit ferromagnetic order such as CeGaGe [7], CeAgGa [8], CePdSb [9], CeRu₂Ge₂ [10], CeRh₃B₂ [11].

In the Ce–Au–Ge system [10,12–24], the equiatomic compound CeAuGe crystallizes in the NdPtSb type structure and orders ferromagnetically at 10 K [14,22] while CeAu₂Ge₂ [17] adopts the ThCr₂Si₂ structure type and orders antiferromagnetically at 16 K [10]. Other notable compounds are Ce₂AuGe₆ (Ce₂CuGe₆ type) [12], Ce₅Au_xGe_{4–x} [$x=0.43$] [24] which has a Sm₅Ge₄ type structure and CeAu_{0.75}Ge_{1.25} a AlB₂ related compound which orders ferromagnetically at 6 K [19].

During our attempts to synthesize Ce₂AuGe₃, we observed the formation of a mixture of AlB₂-type CeAu_{0.5+x}Ge_{1.5–x} and ThSi₂-type CeAu_{0.5–x}Ge_{1.5+x}. This was also noticed by Jones et al. [19] but did not report details about the crystal structure

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and physical properties. A few other ternary systems such as CeAl_xGe_{2–x} [25], CeCu_xGe_{2–x} [26] and CePd_xGe_{2–x} [19] also were reported as mixtures of AlB₂ and ThSi₂ type compounds. The wide compositional homogeneity ranges on these systems make it difficult to obtain pure phases for physical property studies. Here we succeeded in obtaining single phase of ThSi₂-type CeAu_{0.28}Ge_{1.72} using high frequency induction heating and we report the single crystal structure as well as its magnetic and heat capacity properties.

2. Experimental

2.1. Reagents

The following reagents were used as purchased without further purification. Ce (metal chunk, 99.9%, Chinese Rare Earth Information Center, Inner Mongolia, China), Au (pieces, 99.9% Alfa Aesar, Ward Hill, MA), Ge (ground from 2–5 mm pieces 99.999% Plasmaterials, Livermore, CA).

2.2. Synthesis

CeAu_{0.28}Ge_{1.72} was obtained by combining 0.71 mmol of the cerium metal, 0.2 mmol of gold, 1.25 mmol of germanium. The reactants were sealed in tantalum ampoules under an argon atmosphere in an arc-melting apparatus. A total of 230 mg was used for the synthesis. The tantalum ampoules were subsequently placed in a water-cooled sample chamber of an induction furnace (EasyHeat induction heating system, Model 7590), first rapidly heated to ca. 1250 K and kept at that temperature for 10 min.

Finally, the temperature was lowered to 1000 K and the sample was annealed at that temperature for another half an hour, followed by quenching by switching off the power supply. The brittle product could easily be separated from the tantalum tube. No reaction with the container was observed. The polycrystalline sample has a metallic silver color and was used for the elemental analyses, structure characterization, and physical property measurements. More than 95 wt% of the desired compound was obtained.

2.3. Elemental analysis

Quantitative microprobe analyses of CeAu_{0.28}Ge_{1.72} were performed with a Hitachi S-3400 scanning electron microscope (SEM) equipped with a PGT energy dispersive X-ray analyzer. Data were acquired with an accelerating voltage of 25 kV and a 60 s accumulation time. The EDS analysis taken on visibly clean surfaces of the samples gave an atomic composition in good agreement with the results derived from the single crystal X-ray diffraction refinement.

2.4. X-ray crystallography

To determine the phase identity and purity, powder X-ray diffraction patterns of CeAu_{0.28}Ge_{1.72} were collected at RT on a CPS 120 INEL X-ray diffractometer with CuK α radiation, equipped with a position-sensitive detector and were compared to the pattern calculated from the single crystal structure refinement.

The X-ray single crystal intensity data were collected at 100 K using a STOE IPDS 2T diffractometer with graphite-monochromatized MoK α ($\lambda=0.71073$ Å) radiation. The X-AREA (X-RED and X-SHAPE within) package suite was used for the data extraction and integration and to apply empirical and analytical absorption corrections. The atomic coordinates were taken from the ThSi₂ structure and refined with SHELXTL program [27]. A stable refinement for CeAu_{0.28}Ge_{1.72} was accomplished only in the tetragonal space group *I*4₁/*amd*. Data collection and refinement details are given in Table 1. The final atomic positions and

Table 1
Crystal data and structure refinement for CeAu_{0.28}Ge_{1.72}.

Empirical formula	CeAu _{0.28} Ge _{1.72}
Formula weight	320.13
Temperature	293(2) K
Wavelength	0.71073 Å
Crystal system, space group	Tetragonal, <i>I</i> 4 ₁ / <i>amd</i>
Unit cell dimensions	<i>a</i> = 4.2415(6) Å <i>b</i> = 4.2415(6) Å <i>c</i> = 14.640(3) Å
Volume	263.38(7) Å ³
Z, Calculated density	4, 8.073 g/cm ³
Absorption coefficient	51.657 mm ⁻¹
<i>F</i> (000)	541
Crystal size	100 × 100 × 50 μm
Theta range for data collection	5.00° – 29.04°
Limiting indices	–5 ≤ <i>h</i> ≤ 5, –5 ≤ <i>k</i> ≤ 5, –20 ≤ <i>l</i> ≤ 20
Reflections collected / unique	1069 / 111 [R(int) = 0.0759]
Completeness to $\theta = 29.04$	97.4%
Refinement method	Full-matrix least-squares on <i>F</i> ²
Data / restraints / parameters	111 / 0 / 9
Goodness-of-fit on <i>F</i> ²	1.242
Final <i>R</i> indices [<i>I</i> > 2 σ (<i>I</i>)]	<i>R</i> 1 = 0.0532, <i>wR</i> 2 = 0.1127
<i>R</i> indices (all data)	<i>R</i> 1 = 0.0541, <i>wR</i> 2 = 0.1122
Extinction coefficient	0.0000(15)
Largest diff. peak and hole	6.431 and –2.055 e. Å ⁻³

$$R = \frac{\sum ||F_o| - |F_c||}{\sum |F_o|}, \quad wR = \frac{\{\sum [w(|F_o|^2 - |F_c|^2)^2] / \sum [w(|F_o|^4)]\}^{1/2}}{1/(\sigma^2(1) + 0.0016I^2)}$$

Table 2
Atomic coordinates ($\times 10^4$), equivalent isotropic displacement parameters ($\text{Å}^2 \times 10^3$) and anisotropic displacement parameters ($\text{Å}^2 \times 10^3$) for CeAu_{0.28}Ge_{1.72}.

	<i>x</i>	<i>y</i>	<i>z</i>	<i>U</i> ₁₁	<i>U</i> ₂₂	<i>U</i> ₃₃	<i>U</i> _{eq}	Occ.
Ce	0	7500	1250	1(1)	1(1)	2(1)	1(1)	1
Au	0	2500	2919(1)	6(1)	13(1)	7(1)	8(1)	0.14
Ge	0	2500	2919(1)	6(1)	13(1)	7(1)	8(1)	0.86

U(eq) is defined as one third of the trace of the orthogonalized *U*_{*ij*} tensor. The anisotropic displacement factor exponent takes the form: $-2\pi^2[h^2a^2 * U^{11} + \dots + 2hka * b * U^{12}]$. *U*₂₃ = *U*₁₃ = *U*₁₂ = 0.

Table 3
Interatomic distances (Å) in the structure of CeAu_{0.28}Ge_{1.72}, calculated with the powder lattice parameters.

Ce:	7	<i>M</i>	3.2349(15)	<i>M</i> :	1	<i>M</i>	2.435(4)
	4	<i>M</i>			2	<i>M</i>	2.4494(19)
					1	Ce	3.2349(15)
					4	Ce	3.2368(8)

All distances of the first coordination spheres are listed. *M* denotes Au_{0.28}Ge_{1.72}.

equivalent isotropic displacement parameters are given in Table 2, and selected bond distances are listed in Table 3.

2.5. Magnetic measurements

Magnetic susceptibility measurements were carried out with a Quantum Design MPMS SQUID magnetometer. Polycrystalline sample of CeAu_{0.28}Ge_{1.72} was loaded without grinding into gelatin capsules, mounted in a plastic straw and affixed to the end of a carbon fiber rod. Temperature dependence data were collected for both zero field cooled (ZFC) and field cooled mode (FC) between 2 and 300 K, with applied field 1 G. Field dependent magnetic measurements were acquired at 2, 5, and 20 K with field sweeping from –50 up to 50 kG for CeAu_{0.28}Ge_{1.72}. The raw data were corrected for the diamagnetic contribution.

2.6. Specific heat

Heat capacity (*C*) measurements were performed on selected polycrystalline sample of CeAu_{0.28}Ge_{1.72}, by relaxation method using QD-PPMS. The sample was glued to calibrated HC-puck using Apizeon N grease. *C* was measured in the 2–50 K range with no applied magnetic fields (*H*). The electronic and phonon contributions derived by fitting the *C* vs. *T* plot using the Debye function [28] within the temperature range 15–25 K, well above the ferromagnetic ordering.

3. Results and discussion

3.1. Reaction chemistry

CeAu_{*x*}Ge_{1–*x*} was discovered from efforts to synthesize Ce₂AuGe₃ using the metal flux technique [29]. The main products obtained from the flux reactions were the binary CeGe₂ [30] and the equiatomic CeAuGe [15] compound. Reactions using high frequency induction heating gave a mixture of ThSi₂- and AlB₂-type CeAu_{*x*}Ge_{1–*x*} similar to the Jones report [19]. Both phases could easily be identified in the X-ray powder diffraction patterns. Single crystals were selected from the mixture and their crystal

structure was refined. We observed that elemental compositions of crystals with $x \sim 0.4$ or less crystallize in the ThSi_2 type (Table 4) and with x higher than 0.6 crystallize in AlB_2 structure type. We also observed both AlB_2 and ThSi_2 phases in the samples with $0.4 < x < 0.6$. It is worth to notice that the lattice constants are increased with Au content. The ThSi_2 structure is common in intermetallic compounds and often shows a wide range of stoichiometric phases especially in the ternary systems [19,25,26]. After determining the composition range from the selected three different single crystals we targeted the starting composition of $\text{CeAu}_{0.25}\text{Ge}_{1.75}$ for direct synthesis using induction heating and obtained the pure compound as only ThSi_2 type and without any AlB_2 type at least up to the level of the powder XRD detection limit. One of the single crystals from this system was refined as $\text{CeAu}_{0.28}\text{Ge}_{1.72}$. The compound is stable in air and no

decomposition was observed even after several months. Irregular shaped single crystals of this new compound have metallic luster. The elemental analysis of this compound with SEM/EDS gave the composition 34 ± 3 at% Ce : 9 ± 1 at% Au : 57 ± 3 at% Ge which is in excellent agreement with the results obtained from the single crystal refinement.

3.2. Crystal structure

The crystal structure of $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ is shown in the Fig. 1 along the b axis. In the ThSi_2 structure type [31] the Ce atom occupies Th ($\bar{4}m2$) position while the Si ($2mm$) positions are filled with a mixture of Au and Ge atoms. The compound is isostructural with CeGe_2 [30] and is made up of a three-dimensional Ge polyanionic network, with the Ce atoms occupying the voids and channels within it (Fig. 1). The lattice constants are compared with the CeGe_2 in Table 4. The Ge–Ge distances are within the range $2.435(4)\text{\AA}$ and $2.449(2)\text{\AA}$, are comparable to the Ge–Ge distances in $\text{CeCu}_x\text{Ge}_{1-x}$ (2.396\AA) [26] and $\text{CeAl}_x\text{Ge}_{1-x}$ (2.454\AA) [25] compounds but slightly larger than in CeGe_2 (2.355\AA) [30]. The Ge–Ge distances are also comparable to those in elemental Ge indicating the existence of the covalent bonds between the atoms. The shortest distance between the Ce and Ge atoms is $3.235(2)\text{\AA}$ which is common in Ce–Ge based intermetallics [25,32–34]. The coordination polyhedra of Ce and M(Au+Ge) are shown in Fig. 2. The M site has 9-fold coordination and is best described as a tri-capped trigonal prism.

Table 4
Compositions and the lattice parameters of $\text{CeAu}_x\text{Ge}_{2-x}$.

Composition	Lattice constants	Method of refinement	Reference
CeGe_2	$a=b=4.210(2)\text{\AA}$, $c=14.182(5)\text{\AA}$	Single crystal	[30]
$\text{CeAu}_{0.15}\text{Ge}_{1.85}$	$a=b=4.227\text{\AA}$, $c=14.387\text{\AA}$	Powder	This work
$\text{CeAu}_{0.28}\text{Ge}_{1.72}$	$a=b=4.2415(6)\text{\AA}$, $c=14.640(3)\text{\AA}$	Single crystal	This work
$\text{CeAu}_{0.39}\text{Ge}_{1.61}$	$a=b=4.245\text{\AA}$, $c=14.712\text{\AA}$	Powder	This work

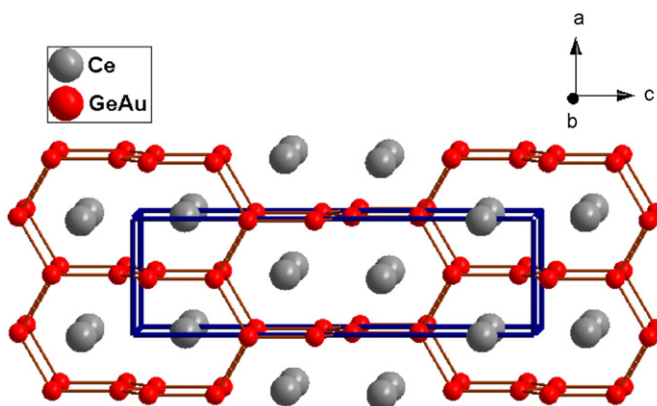


Fig. 1. Crystal structure of $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ along the b -axis.

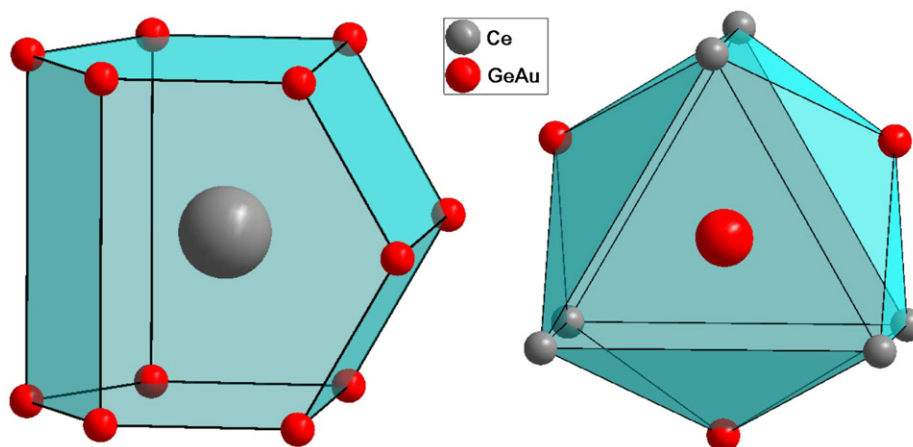


Fig. 2. The coordination polyhedra of Ce and M(Ce+Au) atoms of tetragonal $\text{CeAu}_{0.28}\text{Ge}_{1.72}$.

3.3. Physical properties

3.3.1. Magnetic susceptibility

Fig. 3(a) shows the temperature dependence of the FC and ZFC molar magnetic susceptibility $\chi_m(T)$ of $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ sample. The compound shows a long range ferromagnetic ordering below 8 K and no other magnetic transition down to 2 K. The inset displays the inverse molar magnetic susceptibility $1/\chi_m(T)$ data vs. the temperature. Above 50 K, $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ exhibits paramagnetic behavior and $\chi_m(T)$ nicely follows the Curie–Weiss (CW) law $\chi(T) = C/(T - \theta_p)$ [35], where $C = N_A \mu_{\text{eff}}^2 / 3k_B = \mu_{\text{eff}}^2 / 8$ is the Curie constant and θ_p is the Weiss temperature. Linear least-squares fit to this equation of the data above 100 K resulted in a Weiss temperature of $\theta_p \approx 34\text{K}$ suggesting ferromagnetic interactions between the Ce atoms, and an effective moment of $2.48 \mu_B/\text{Ce}$ atom. This experimental value is close to that expected for a free Ce^{3+} ion ($2.54 \mu_B$). On the other hand, below 50 K the $\chi_m(T)$ data do not obey the CW law, while below $\sim 8\text{K}$ there is a sudden

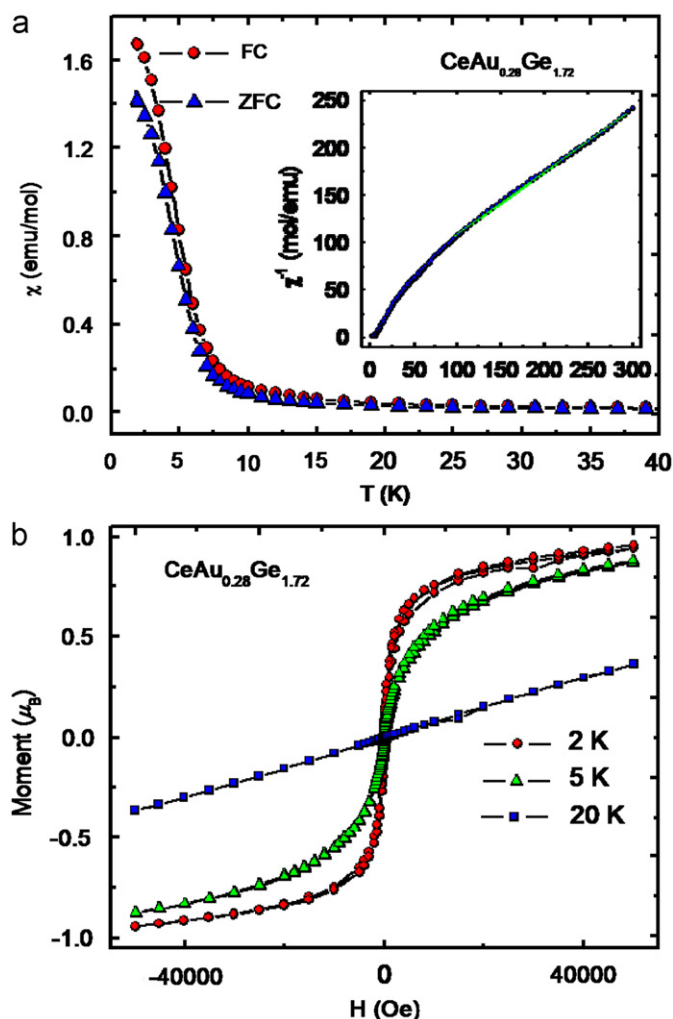


Fig. 3. (a) Field cooled (FC) and zero field cooled (ZFC) magnetic susceptibility ($\chi = M/H$) as a function of temperature for $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ (polycrystalline) sample measured in a dc field of 1 kOe. Inset shows the temperature dependent inverse susceptibility of $\text{CeAu}_{0.28}\text{Ge}_{1.72}$. The linear region is marked clearly in the figure. (b) $M(H)$ measured for $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ polycrystalline sample at various temperatures spanning T_c .

rise in the magnetic moment indicative of ferromagnetic order. The isostructural parent compound CeGe_2 shows both antiferromagnetic ($T_N = 7$ K) and ferromagnetic ordering ($T_c = 4.3$ K) at 1000 Oe [36]. The partial substitution of gold in the germanium position suppresses the antiferromagnetic ordering and shifts the ferromagnetic transition to higher temperatures.

Fig. 3(b) illustrates the field dependent magnetization data for $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ measured at 2, 5 and 20 K. The data measured at 20 K (above the ferromagnetic ordering) exhibit linear behavior and no signs of saturation up to our highest attainable field of 50 kG. The magnetization curve taken at 5 K (within the ferromagnetic ordering) shows a strong field dependent response up to ~ 5.2 kOe, which almost saturates at ~ 20 kOe, but continues to rise slowly up to the highest obtainable field (50 kG). At 2 K the strong field dependent on the magnetic moment up to 5 kOe but the system saturates rapidly.

Inside the ordered region (Fig. 3(b)) the magnetization tends to saturate at slightly less than $1 \mu\text{B}/\text{mol}$ Ce at ~ 25 kG. In spite of the paramagnetic effective moment at higher temperature indicating an unperturbed Ce^{3+} ionic species, the saturation moment is significantly below the theoretical value for Ce ($2.54 \mu\text{B}/\text{mol}$). This may arise from crystal field splitting of the $J = 5/2$ multiplet

ground state of the Ce^{3+} ion in this compound as for example in CeAuGe [16] and CeAgAl_3 [37], which also show a low saturation moment.

3.3.2. Heat capacity

The temperature dependent specific heat from 1.8 to 50 K for $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ is shown in Fig. 4(a). There is a rise in C_p at 8 K confirming the magnetic order in this compound. The temperature where the rise in C_p is observed is close to the Curie temperature of 8 K. There is a peak at 4 K, below which the heat capacity drops. The data can be described well by the Debye function [28] in Eq. (1) where the first and second terms correspond to the electronic and the phonon contribution, respectively. N is the number of the atoms in the formula unit and $\chi = \hbar\omega/k_B T$.

$$C_p(T) = \gamma T + 9NR \left(\frac{T}{\Theta_D} \right)^3 \int_0^{\Theta_D/T} \frac{x^4 dx}{(e^x - 1)^2} - KT^3 \quad (1)$$

A fit to the experimental points could only be made within the temperature region 15–25 K, well-above the ferromagnetic ordering and within the paramagnetic state of the material. The fit resulted in a Debye temperature Θ_D of ~ 184 K and an electronic specific heat coefficient $\gamma \approx 124(5) \text{ mJ}/\text{mol}\cdot\text{K}^2$. Because of the inability to include the 2–15 K data in the fit, the estimate γ

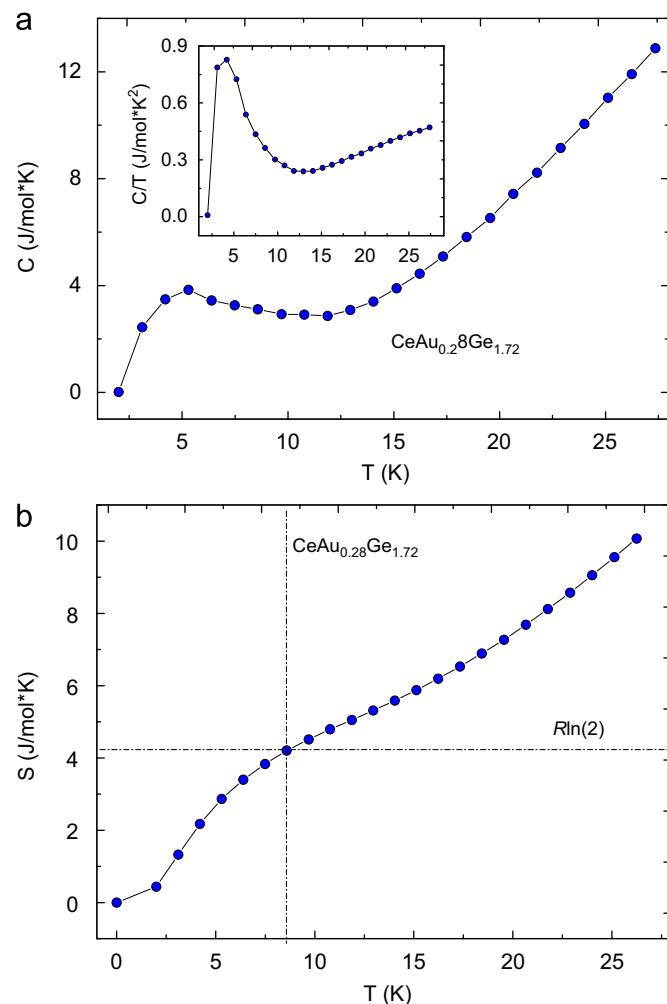


Fig. 4. (a) Heat capacity (C) for $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ measured as a function of temperature (T) at zero applied fields. The inset figure is C/T vs. T . (b) The total entropy (S) vs. T for $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ calculated from C measured at zero applied field. The vertical line indicates the ferromagnetic ordering temperature ($T_N = 8$ K) whereas the horizontal line is drawn to show the value of full magnetic entropy for Ce ($R \ln(2j+1)$ for $J = 1/2$).

value may be subject to a large uncertainty. Ferromagnetic correlations are expected to be small in this temperature range since the lower limit of this temperature (15 K) is close to $\sim 2T_c$. The apparently large value of γ could originate from crystal field effects and/or Kondo interactions.

The entropy associated with the magnetic ordering is given by $\Delta S = \int C_p/T dT$. In Fig. 4(b), we show the total entropy for $\text{CeAu}_{0.28}\text{Ge}_{1.72}$, calculated from the C_p/T vs. T plot. It is clear that the total entropy changes at 8 K in accord with the ferromagnetic ordering. For cerium (S -state ion with zero angular momentum and $J=1/2$) the full magnetic entropy ($R\ln 2$) should be recovered just above the ordering temperature. For $\text{CeAu}_{0.28}\text{Ge}_{1.72}$, the total entropy (S) is not saturated above T_c and increases up to the highest temperature measured, i.e., 50 K without any sign of saturation.

The total entropy gives a value of 4.2 J/mole K over the temperature 7–9 K, where the magnetic ordering occurs. The ΔS obtained is low compared to the value $R\ln(2)=5.75$ J/mol K expected assuming that the Ce ions are in the crystal field split doublet ground state. It is possible that the magnetic entropy is tied up with the magnetic short range order persisting well above the transition temperature in the paramagnetic region. Reduced values of magnetic entropy can also arise from the presence of significant Kondo exchange interactions which may partially quench the local spin [38–40]. Smaller entropy values were also observed for $\text{Ce}_x\text{AlGe}_{2-x}$ [25].

4. Concluding remarks

Single phase ThSi_2 -type $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ was successfully prepared and was found to order ferromagnetically at ~ 8 K. Above this temperature $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ obeys Curie Weiss behavior and Ce is in trivalent state. The ferromagnetic interaction between the Ce atoms is also evident in the rise of the heat capacity observed at ~ 8 K. The estimated enhanced γ (~ 124 mJ/mol K²) value in the paramagnetic range of $\text{CeAu}_{0.28}\text{Ge}_{1.72}$ could indicate possible heavy fermion behavior or crystal field effect and/or Kondo interaction and needs to be investigated further.

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